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# Ferromagnetism and atomic short range order in dilute PdMn alloys

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Abstract. Polarised neutron scattering and magnetisation measurements on dilute PdMn alloys have been analysed with a local environment model of ferromagnetic interactions and atomic short range order. The analysis allows us to calculate several magnetic properties of PdMn such as exchange constants of ferromagnetism, range of magnetisation clouds and chemical short range order parameters. Critical temperatures of ferromagnetism, which are low in dilute PdMn alloys compared to those in other typical giant moment systems such as PdFe, were calculated using the Monte Carlo simulation method and the results, which included atomic short range order, were compared with experimental data. The comparison shows a good agreement between experimental and calculated values for the concentrations considered.

#### 1. Introduction

Giant moments are usually present in metallic alloys with Fe or Co as the impurities. The solvent is generally a strongly exchange enhanced material such as Pd or Pt. It is known, however, that **Pd**Mn is also a ferromagnet at dilute concentrations of the impurity, with an expected enhancement of the Pd moment. The magnetisation work of Star *et al* (1975), showed that Mn has giant moments of about 7.5  $\mu_B$  in Pd with approximately 4.0  $\mu_B$  located in Mn and the rest of the giant moment distributed among the atoms. The polarised neutron scattering data of Cable and David (1977), however, do not show these giant moments. It has also been shown that critical temperatures are much lower in **Pd**Mn than in other dilute giant moment systems (Star *et al* 1975).

We have developed a model (Medina and Parra, 1982), that has been applied successfully to other giant moment systems such as **PdFe**, **PtFe** and **PtCo**. This model has allowed us to predict critical temperatures, magnetisation and several other magnetic properties of these materials. In this paper we apply this model to study the interesting magnetic behaviour of **Pd**Mn described above, taking into account atomic short range order and assuming a ferromagnetic polarisation of Pd atoms due to the moment of the impurity.

### 2. Model and analysis

We assume a model where atomic short range order (ASRO) and ferromagnetic Pd–Mn interactions are present at all concentrations, becoming increasingly important as the

Mn concentration is augmented. We therefore analysed the neutron scattering results m(K) of Cable and David (1977) for **Pd**Mn in the concentration region  $0.23 \le c \le 1.91$  at % Mn with a function:

$$m(\mathbf{K}) = M(\mathbf{K})S(\mathbf{K}) \tag{1}$$

where S(K) is the usual ASRO function:

$$S(\mathbf{K}) = \sum_{j=0}^{n} Z_j a(\mathbf{R}_j) e^{i\mathbf{K} \cdot \mathbf{R}_j}$$
(2)

where we sum over the *n* near neighbour shells, with  $Z_j$  as the coordination numbers and  $a(R_j)$  as Cowley's ASRO parameters. The relation between the ASRO parameters and the probabilities *P* of finding an impurity atom as a *j* neighbour of a matrix atom is  $a_j = (1 - P_j/c)$ , and the number *N* of Mn atoms as a *j* neighbour of Pd atoms is given by  $N_j = P_j Z_j$ .

To analyse the ferromagnetic correlations M(K), we apply a magnetic environment model to the polarised neutron scattering data. In this model (Medina and Parra 1982) the moment of a Pd atom at site n is given by

$$\mu_n = \chi_0 h_n \tag{3}$$

where  $\chi_0$  is the non-enhanced susceptibility and  $h_n$  is a local field given by

$$h_{n} = b_{n} - J_{\text{Pd}} \sum_{\delta} \mu_{n+\delta} [1 + (\alpha - 1)P_{n+\delta}].$$
(4)

Here  $b_n$  is the external field,  $J_{Pd}$  is the exchange constant between Pd atoms,  $\alpha J_{Pd}$  is the exchange constant between Mn atoms and  $P_n$  is a site occupation operator.

With no impurities present we obtain from (3) and (4)

$$\chi = \chi_0 / (1 - \Gamma) \tag{5}$$

where  $\Gamma = J_{Pd}Z_1\chi_0$ . The parameter  $\Gamma$  is the susceptibility enhancement parameter of Pd and was determined by Parra and Medina (1980) as  $\Gamma = 0.947 \pm 0.009$ , from experimental neutron data. Using the reported value of  $\chi = 6.85 \ 10^{-6} \ \text{emu g}^{-1}$  (Verbeek *et al* 1980) and (5), Medina and Parra (1982) obtained  $\chi_0 = 3.63 \ 10^{-7} \ \text{emu g}^{-1}$  and  $J_{Pd} = (766 \pm 100) \ \text{K}$ .

The Fourier transform of (3) gives us an expression for the magnetisation cloud that can be related to the neutron scattering data:

$$M(\mathbf{K}) = \mu_{\rm Mn} f_{\rm Mn}(\mathbf{K}) [\![\alpha / \{\varphi_0 [1 - \Gamma F_1(\mathbf{K})]\} + (1 - \alpha)]\!]$$
(6)

where  $F_1(\mathbf{K})$  is the first neighbour shell structure factor and  $\varphi_0$  is given by

$$\varphi_0 = \frac{1}{V_c^*} \int \frac{\mathrm{d}^3 K}{1 - \Gamma F_1(K)}$$
(7)

which were determined by Medina and Parra (1982). To be able to use this equation we needed the values of  $\mu_{Mn}$  for each concentration. These were obtained by fitting the neutron data of Cable and David (1977) with the following equations, valid for polarised neutrons:

$$m(\mathbf{K}) = \mu_{\mathrm{Mn}} f_{\mathrm{Mn}}(\mathbf{K}) - \mu_{\mathrm{Pd}} f_{\mathrm{Pd}}(\mathbf{K})$$
(8)

$$\mu = \mu_{\rm Mn} c + (1 - c) \mu_{\rm Pd}.$$
(9)

**Table 1.** Values of individual average magnetic moments of Mn and Pd, first neighbour moment perturbations (calculated using (12)) and range parameter of magnetisation clouds in dilute **PdMn** alloys.

c (at.% Mn)	$\mu_{ m Mn}$	$\mu_{ m Pd}$	$oldsymbol{arphi}_1$	$K_0$
0.23	3.58	7.78 10 <sup>-3</sup>	3.63 10-2	0.2010
0.46	4.14	$1.10\ 10^{-2}$	$3.10 \ 10^{-2}$	0.2031
0.99	3.52	$2.34 \ 10^{-2}$	3.71 10-2	0.2008
1.91	3.62	$3.04 \ 10^{-2}$	$3.71 \ 10^{-2}$	0.2056
	±0.07	$\pm 0.02 \ 10^{-2}$	$\pm 0.04 \ 10^{-2}$	$\pm 0.0001$

In these expressions we used  $f_{Mn}(K) = \exp(-0.069 K^2)$  and  $f_{Pd}(K) = \exp(-0.1 K^2)$ , which closely approximate the experimental form factors, and  $\mu$  values obtained from magnetisation data. The results for  $\mu_{Mn}$  and  $\mu_{Pd}$  are given in table 1 for each concentration. With these values we can obtain  $J_{PdMn}$  knowing the total moment of the cloud M(0). Star *et al* (1975) report several values for M(0) in this concentration range. We used an average value of  $M(0) = 7.12 \mu_B/Mn$  in this range. Using (6) we then calculate  $J_{PdMn}$  obtaining an average value of  $J_{PdMn} = (48 \pm 7) \,^{\circ}$ K.

According to this model, calculation of M(K) using (6), should give us the experimental neutron m(K) values for a ferromagnet. We would now like to obtain more information on the ferromagnetic interactions of this alloy and the magnetisation cloud associated with it. We therefore fit these calculated values using Marshall's (1968) model equation:

$$M(\mathbf{K}) = \mu_{\mathrm{Mn}} f_{\mathrm{Mn}}(\mathbf{K}) - \mu_{\mathrm{Pd}} f_{\mathrm{Pd}}(\mathbf{K}) + \sum_{i} Z_{i} \varphi_{i} \frac{\sin K R_{i}}{K R_{i}}$$
(10)

where the  $\varphi_i$  have the form of a Yukawa potential

$$\varphi_i = (R_1/R_i)\varphi_1 \exp[-K_0(R_i - R_1)].$$
(11)

In this equation  $K_0$  represents the range of the magnetisation clouds and the  $\varphi$  values are the moment perturbations. Substituting (11) into (10) we obtain the following fitting function:

$$M(\mathbf{K}) = \mu_{\rm Mn} f_{\rm Mn}(\mathbf{K}) - \mu_{\rm Pd} f_{\rm Pd}(\mathbf{K}) + \sum_{i} Z_{i} \frac{R_{1}}{R_{i}} \varphi_{1} \frac{\sin KR_{i}}{KR_{i}} \exp[-K_{0}(R_{i} - R_{1})].$$
(12)

The fitting of the data was made by a non-linear least squares method with parameters  $\varphi_1$  and  $K_0$ . The results are shown in table 1, and a comparison of the fitting with calculated M(K) values and the experimental m(K) is presented in figure 1. Since M(K) comes from a ferromagnetic model, it is noticeable that for c = 0.23 at.% Mn, ferromagnetic interactions are predominant, while at c = 1.91 at.% Mn ASRO plays an important part in the neutron scattering data.  $K_0$  is the width of the peak at small K, and its average value for these concentrations is  $K_0 = 0.203$ , which indicates the presence of magnetisation clouds with a range of 15.5 Å due to ferromagnetic interactions.

A fitting of  $m(\mathbf{K})$  for a polycrystal to equation (1) is shown in figure 2, where  $M(\mathbf{K})$  are the values calculated using equations (6) and (12), and  $S(\mathbf{K})$  is given by (2). Values



**Figure 1.** Experimental neutron scattering data from Cable and David (1977) are compared with the values obtained from the model (theory) and from the fitting (M(K)) of those values obtained using (12).

of the  $a(R_j)$  parameters are presented in table 2. These values tend to alternate in sign as expected, the first parameter being negative, indicating the preference of the impurity to have matrix atoms as first neighbours. They also show an increasing importance of this order as the concentration of Mn is raised. The fitting with these equations describe quite well the neutron scattering data.

ASRO has been found in Pd-9.95 at.% Mn by Morgownik *et al* (1987) and its importance for the spin glass behaviour of **PdM**n at this concentration has been shown by those authors. Their parameters are compared with ours in table 3 and show good agreement.

The model that we are using also allows the calculation of the critical temperature of ferromagnetism. This is important because **PdM**n shows critical temperatures for the onset of ferromagnetism that are roughly ten times smaller than the critical temperatures of **PdFe**, a typical giant moment system, for the same concentration. We then made Monte Carlo calculations for c = 0.23, 0.46, 0.99 and 1.91 at.% Mn, and considered a system of classical spins obeying the following Hamiltonian:

$$H = (-1/2) \sum_{\substack{i,j\\i \neq j}} J_{ij}(R_{ij}) \mathbf{S}_i \cdot \mathbf{S}_j$$
(13)

where the values of  $J_{ij}$  were calculated by Medina and Parra (1982). For each concentration we used samples of 100 spins with 2000 Monte Carlo steps per spin. We calculated, for several temperatures, the energy E, the z component and the magnitude of the magnetic moment M, the susceptibility  $\chi$  (from the average fluctuation of a component of M) and a quantity  $\tilde{\chi}$  which is given by the fluctuation of the magnitude of the total moment and which is equal to  $\chi$  for an infinite system.  $\tilde{\chi}$  is given by

$$\tilde{\chi} = \frac{\langle (M - \langle M \rangle)^2 \rangle}{NkT} \tag{14}$$

where N is the number of spins. Critical temperatures were obtained from the maximum



Figure 2. Neutron scattering m(K) from Cable and David (1977) compared with results obtained from the model. M(K) is the ferromagnetic part of the model and S(K) is the atomic short range order.

c (at.% Mn)	a(0)	$a(R_1)$	$a(R_2)$	$a(R_3)$
0.23	1.007(17)	-0.016(6)	0.017(28)	-0.0008(60)
0.46	0.988(9)	-0.017(3)	0.032(15)	-0.0087(30)
0.99	0.985(6)	-0.023(2)	0.043(9)	-0.0168(20)
1.91	0.966(3)	-0.031(1)	0.044(5)	-0.0214(10)
9.95	1.000	-0.068(3)	0.027(6)	-0.0030(10)

**Table 2.** Atomic short range order parameters of dilute **PdM**n alloys, as calculated with the model. The ASRO parameters for a 9.95 at.% Mn, obtained by Morgownick *et al* (1987), are also shown.

of  $\tilde{\chi}$  against temperature. To compare the calculated critical temperatures with the experimental ones (in kelvin) one must in general multiply the calculated  $T_c$  values by a factor  $J_0 = J_{Pd} (\alpha \mu_{Mn})^2$  which comes from (6). Notice, however, that we are analysing

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**Table 3.** Comparison of the number  $N_j$  of Mn atoms as *j* neighbours of Pd atoms, using our calculated ASRO parameters on a 0.99 at.% Mn and the ones obtained by Morgownick *et al* (1987) on a 9.95 at.% Mn sample, with approximately the same heat treatment. N for the 9.95 at.% Mn sample is nearly 10 times larger than N for the 0.99 at.% Mn sample, as expected.

$N_1$	$N_2$	$N_3$
0.1215	0.0569	0.2416
1.2752	0.5809	2.3808
	N <sub>1</sub> 0.1215 1.2752	$ \begin{array}{cccc} N_1 & N_2 \\ \hline 0.1215 & 0.0569 \\ 1.2752 & 0.5809 \\ \end{array} $



Figure 3. Experimental critical temperatures of ferromagnetism from Star *et al* (1975) compared with the ones obtained by Monte Carlo calculations using the model.

the neutron scattering m(K) with a function M(K)S(K) and therefore our factor  $J_0$  must be:

$$J_0 = J_{\rm Pd} [\alpha \mu_{\rm Mn} S(0)]^2 \tag{15}$$

where S(0) is the calculated value of S(K) at K = 0. In this way we are taking into account the ASRO. The  $T_c$  values obtained this way are compared in figure 3 with experimental values. We observe that the calculated values agree, in general, with experimental  $T_c$  values.

## 3. Conclusions

Two problems are remarkable concerning the ferromagnetism of dilute PdMn alloys. First, magnetisation measurements by Star *et al* (1975) showed that giant moments could be measured in PdMn alloys, just as in PdFe or in PdCo, which are typical giant moment systems. However, polarised neutron scattering data (Cable and David 1977) on PdMn do not show the giant moments found by magnetisation measurements. The other problem is that PdMn was very different to other giant moment systems regarding the onset of ferromagnetism which occurs in PdMn at much lower critical temperatures. In this paper we have presented an analysis of PdMn data with a magnetic environment model which, with simple assumptions, establishes a compatibility between neutron and magnetisation measurements. The model also allows the calculation, by Monte Carlo simulation, of the critical temperatures of ferromagnetism which agree well with experimental ones. In this model, for very low concentrations ferromagnetic interactions dominate making possible the formation of giant moments. As the Mn concentration is increased ASRO becomes important. This introduces a preference of Pd atoms as first neighbours of Mn atoms and this opposes the long range ferromagnetic order, which up to now was attributed to antiferromagnetic interactions among Mn atoms. The coexistence of these two characteristics makes possible their measurement by different experimental techniques. The presence of ASRO forces critical temperatures to be lower than in other giant moment systems where only ferromagnetic interactions are present.

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